

# Spatial and spectral detection of protein monolayers with deterministic aperiodic arrays of metal nanoparticles

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**Light scattering phenomena in periodic systems have been investigated for decades in optics and photonics. Their classical description relies on Bragg scattering, which gives rise to constructive interference at specific wavelengths along well defined propagation directions, depending on illumination conditions, structural periodicity, and the refractive index of the surrounding medium. In this paper, by engineering multifrequency colorimetric responses in deterministic aperiodic arrays of nanoparticles, we demonstrate significantly enhanced sensitivity to the presence of a single protein monolayer. These structures, which can be readily fabricated by conventional Electron Beam Lithography, sustain highly complex structural resonances that enable a unique optical sensing approach beyond the traditional Bragg scattering with periodic structures. By combining conventional dark-field scattering micro-spectroscopy and simple image correlation analysis, we experimentally demonstrate that deterministic aperiodic surfaces with engineered structural color are capable of detecting, in the visible spectral range, protein layers with thickness of a few tens of Angstroms.**

biosensing | colorimetric fingerprint | deterministic aperiodic system | nanophotonics

In current biosensing technology, two-dimensional periodic lattices, (i.e., two-dimensional optical gratings) provide a well established approach for biochemical colorimetric detection, which can yield label-free sensing of various molecular analytes and protein dynamics (1–7). Standard periodic grating biosensors provide a distinct change either in the intensity of diffracted light or in the frequency of optical resonances in response to changes in the refractive index of the surrounding environment (8).

The physical mechanism at the base of these optical signatures is the well known phenomenon of Bragg scattering. While this process provides frequency selective responses that are useful for colorimetric detection, the ability of light waves to interact with adsorbed or chemically bound analytes present on the surface of these sensors is intrinsically limited. In fact, Bragg scattering is a first-order process in surface scattering perturbation theory (9, 10), and scattered photons easily escape from a periodic surface within well defined spectral bands and without prolonged interaction with the sensing layer. In this paper, we present an approach to overcome this fundamental limitation by developing a label-free biosensing approach based on micro-spectroscopy and spatial correlation imaging (11) of structural color patterns obtained by white light scattering from two-dimensional nanoscale deterministic aperiodic structures.

The proposed concept relies on the specific optical properties of deterministic aperiodic photonic surfaces, which lack translational invariance symmetry (they are nonperiodic) and are generated by simple constructive rules (12, 13). Such structures, which can be fabricated using conventional lithographic techniques, are an intermediate regime between periodic and disordered systems, yet are engineered according to mathematical

rules amenable to predictive theories. In contrast to traditional photonic gratings or photonic crystals sensors (which efficiently trap light in small-volume defect states), aperiodic scattering sensors sustain distinctive resonances localized over larger surface areas. In particular, nanoscale aperiodic structures possess a dense spectrum of highly complex structural resonances, called critical modes, which result in efficient photon trapping and surface interactions through higher-order multiple scattering processes thereby enhancing the sensitivity to refractive index changes (14, 15). The complex spatial patterns of critical modes in these structures offer the potential to engineer structural color sensing with spatially localized patterns at multiple wavelengths, which we call colorimetric fingerprints.

Here we propose to utilize critical mode patterns as surface sensing elements with sensitivity to protein monolayer morphological changes. The proposed approach is intrinsically more sensitive to local refractive index modifications compared to traditional ones (14) due to the enhancement of small phase variations, which is typical in the multiple light-scattering regime (9, 10). By combining Electron Beam Lithography (EBL), dark-field scattering micro-spectroscopy, and rigorous calculations based on the Generalized Mie Theory (GMT) (16), we explain the origin of structural color localization in aperiodic arrays of Chromium (Cr) nanoparticles on quartz substrates and demonstrate their potential for the detection of protein monolayers. In addition, we show that the complex spatial patterns of critical modes in nanostructured aperiodic surfaces are ideally suited for image correlation analysis in the visible spectral range, providing a transduction mechanism with large dynamic range, sensitivity and multiplexing capabilities where the information encoded in *both* spectral and spatial distributions of structural colors can be simultaneously utilized. The potential of the proposed approach for rapid, label-free detection of biomolecular analytes performed in the visible spectral range using a commercially available microscope is experimentally demonstrated by showing a distinct variation in the spectral and spatial colorimetric fingerprints in response to monolayer increments of protein layers sequentially deposited on the surface of aperiodic arrays of nanoparticles.

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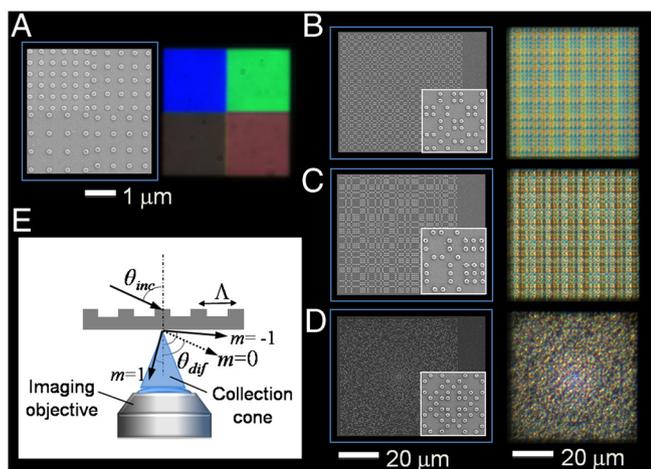
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## Results and Discussion

**Colorimetric Fingerprints of Periodic and Aperiodic Gratings.** Before discussing light scattering from aperiodic nanopatterned surfaces, we will briefly review the colorimetric response of periodic arrays of Cr nanoparticles deposited on quartz substrates. Two-dimensional periodic gratings of 100 nm-radius and 40 nm-tall Cr nanodisks (shown in Fig. S1) of varying lattice constants were fabricated on quartz substrates using EBL (detailed in *Materials and Methods*) and the scanning electron micrographs of representative grating structures are shown in Fig. 1A. The arrays are illuminated by an incoherent white light source at a grazing angle incidence ( $\theta_{\text{inc}} = 75$  degrees) to the array surface using the dark-field scattering setup sketched in Fig. 1E, and a microscope objective lens is used to collect the scattered radiation normal to the array plane (detailed in *Materials and Methods*). Increasing the grating period results in a progressive red-shift of the colorimetric responses (scattered wavelengths), as shown by the experimental data in Fig. 1A. These colorimetric responses of periodic gratings can adequately be described by the classical Bragg formula:

$$\lambda = \frac{\Lambda}{m} (n_1 \sin \theta_{\text{inc}} \pm n_2 \sin \theta_{\text{dif}}), \quad m = 0, \pm 1, \pm 2, \dots \quad [1]$$

where  $\Lambda$  is the lattice constant,  $\lambda$  is the wavelengths of the incident light,  $\theta_{\text{inc}}$  and  $\theta_{\text{dif}}$  are the incident and the diffracted angles (measured with respect to the normal to the grating surface),  $m$  is the order of diffraction and  $n_1$  and  $n_2$  are the refractive indices of the grating and of the surrounding medium, respectively. The calculated colorimetric responses corresponding to different lattice constants (500 nm–800 nm) vary as a function of the diffraction angle. In addition, the spectral response is determined by the finite angular collection efficiency of the imaging lens as depicted in Fig. 1E by the blue area. The distinctive wavelength shift of the radiation scattered by periodic gratings perturbed by the presence



**Fig. 1.** Colorimetric fingerprints of periodic and aperiodic gratings. (A–D) SEM images of two-dimensional periodic and aperiodic arrays of 100 nm-radius and 40 nm-high cylindrical Cr nanoparticles on a quartz substrate and the associated dark-field images were illuminated at a grazing incidence with white light. The structural color patterns of the images vary by the N.A. of the imaging objective, in which different diffractive order is included into the collection cone. The periodic arrays in (A) were observed under 10 $\times$  objective with a 1 mm iris of N.A. reduced to 0.1 and the aperiodic arrays in (B) Thue-Morse lattice (nearest center-to-center separation  $\Lambda = 400$  nm); (C) Rudin-Shapiro lattice ( $\Lambda = 400$  nm); (D) Gaussian prime lattice ( $\Lambda = 300$  nm), were observed under 50 $\times$  objective with N.A. 0.5. The structural color patterns also vary by increasing the grating period with a progressive red-shift of the scattered wavelengths in (A) (clockwise from top-left) (E) A schematic of the dark-field scattering setup used in the measurements.

of specific analytes has been traditionally utilized as a transduction signal in colorimetric optical sensing (3–7).

Unlike periodic grating structures, the scattering response of aperiodic nanopatterned surfaces features highly complex colorimetric fingerprints, as demonstrated in Fig. 1B–D. We focus here on three main types of deterministic aperiodic structures with varying degree of structural disorder. Specifically, we focus on Thue-Morse (12–15, 17) (Fig. 1B), Rudin-Shapiro (12–15, 18) (Fig. 1C), and Gaussian prime (19) (Fig. 1D) arrays of Cr nanoparticles with minimum center-to-center separation of 300 nm and 400 nm. The spatial complexity of these aperiodic structures is described by the spectral character of their spatial Fourier spectra, which in contrast to simple periodic structures, densely fills the reciprocal space with distinctive fractal properties (19–22). In particular, Gaussian prime lattices feature nonperiodic Fourier spectra with well defined reciprocal lattice vectors (Bragg-peaks) (19), while the more complex Thue-Morse and Rudin-Shapiro structures display singular continuous and absolutely continuous Fourier spectra (12–15, 17, 18), respectively. All these aperiodic surfaces possess a large number of spatial frequencies, which can assist higher-order in-plane scattering processes and excite the critical resonances of the systems.

When these structures are illuminated by a white light source, they give rise to highly organized structural color patterns as shown in Fig. 1. (Additional patterns obtained from different deterministic aperiodic arrays are shown in Fig. S2). In the next section, the origin of the experimentally observed colorimetric fingerprints is explained using rigorous multiple scattering theory on model structures in three spatial dimensions.

**Structural Color Formation in Aperiodic Structures.** It is well known that aperiodic systems possess a dense spectrum of critical modes, featuring unique fractal scaling and spatial localization character with traits intermediate between Anderson and Bloch modes (15, 20, 21). When these modes are excited, photons can be efficiently trapped on the surface of aperiodic systems enabling enhanced surface interactions in comparison to what can be achieved using traditional optical modes (14).

The origin of the experimentally observed colorimetric fingerprints of aperiodic arrays of subwavelength particles was analyzed by performing three dimensional light-scattering simulations on model structures consisting of Cr nanospheres arranged in periodic and aperiodic two-dimensional lattices. These structures are illuminated by a plane wave incident at a grazing angle ( $\theta_{\text{inc}} = 75$  degrees- consistent with our experimental conditions) to the array plane. The far-field scattering characteristics and intensity distribution in the array plane of the scattered electric field have been calculated using the rigorous GMT approach (16, 23) (further details on the GMT method appear in the *SI Text*).

The formation of this distinctive multispectral response is illustrated in Fig. 2A–D for the case of Gaussian prime arrays. The calculated scattering spectrum of the Gaussian prime array (Fig. 2E) illuminated by a plane wave reveals variations of the array scattering efficiency (the ratio of the scattering cross section to the total volume of the particles (13)) as a function of the wavelength. Furthermore, the calculated scattered intensity pattern in the plane of the array features different spatial distributions of critical modes corresponding to different wavelengths (Fig. 2A–C). When the colorimetric patterns of the Red-Green-Blue (RGB) principal chromatic components (wavelengths 630 nm, 520 nm, and 470 nm) are mixed together in the array plane (Fig. 2D), a complex structural color pattern (colorimetric fingerprint) is formed in qualitative agreement with the experimentally measured data, shown in Fig. 2F, collected under white light illumination.

The formation of this complex pattern illustrates the possibility of spatial localization of individual frequency components on the nanostructured surface. Due to the aperiodicity of the structure,



ing spectrum measured in the presence of protein layers (Fig. 3E) is quantified by estimating the slope of the Peak Wavelength Shift (PWS) plotted versus the thickness of the protein layer. A linear fit of the experimental data shown in Fig. 3F demonstrates device sensitivity of approximately 1.5 nm per protein monolayer (~20 Angstroms). This value is comparable to that presently reported for photonic crystal structures and surface plasmon biosensors (5, 25, 26). The smallest detection volume of silk protein is estimated as  $A(t)(D/M)$ , where  $A$  is the total surface area of the Gaussian prime nanopatterned array ( $48.2 \times 48.2 \mu\text{m}^2$ ),  $t$  is the film thickness (2 nm),  $D$  is the density of the protein ( $1.4 \text{ g/cm}^3$ ) (27), and  $M$  is the molecular mass of the protein (375 kDa) (28). About 17 atto-mole of protein molecules is estimated in our experiment contributing to the distinctive shift of the spectral peak and the colorimetric pattern change. We note that this detection limit can be improved by minimizing the size of the nanopatterned surface. It is important to note that no protein detection can be observed in the 2–5 nm thickness range when using periodic gratings with Bragg scattering efficiency optimized in the same spectral region as the Gaussian prime arrays. As shown in Fig. 4, periodic grating sensors excited in the same experimental geometry do not reveal any spectral shift in response to the deposition of 2–5 nm thick protein layers on the surface of the samples. A small colorimetric response is detected when 20 nm thick layers are deposited on the periodic gratings, corresponding to a small shift in the peak of their scattering spectra (Fig. 4C). Enhanced sensitivities using periodic gratings can only be achieved by measuring enhanced backscattering intensities or by introducing structural defects to form photonic crystal cavities at specific wavelengths (3, 5).

On the other hand, aperiodic surfaces with engineered colorimetric fingerprints additionally offer the possibility to detect protein monolayers by observing, with conventional dark-field microscopy, distinctive structural modifications of the spatial distribution of the individual spectral components of the scattered radiation field, as demonstrated in Fig. 5A–D in the case of silk nanolayers. This detection mechanism takes advantage of the fingerprinting structural resonances perturbed by the presence of nanoscale protein layers. Therefore, in the case of aperiodic structures, both the peak wavelength shift of the scattered radiation as well as the spatial structure of their distinctive colorimetric fingerprints can be utilized in order to detect the

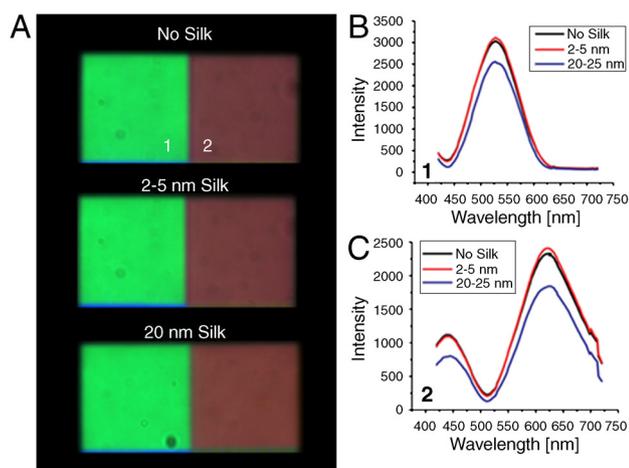
presence of nanoscale protein layers. The spatial modifications of the structural color fingerprints of aperiodic surfaces can be readily quantified by image autocorrelation analysis performed on the radiation intensity scattered by the bare surface and by the silk coated surface (29, 30). The two-dimensional image autocorrelation function (ACF) of a colorimetric fingerprint  $G(\xi, \eta)$  can be obtained directly from the scattering data by proper normalization as (detailed in the *SI Text*):

$$g(\xi, \eta) = \langle \delta s(x, y) \delta s(x + \xi, y + \eta) \rangle = \frac{G(\xi, \eta)}{\langle s(x, y) \rangle^2} - 1, \quad [2]$$

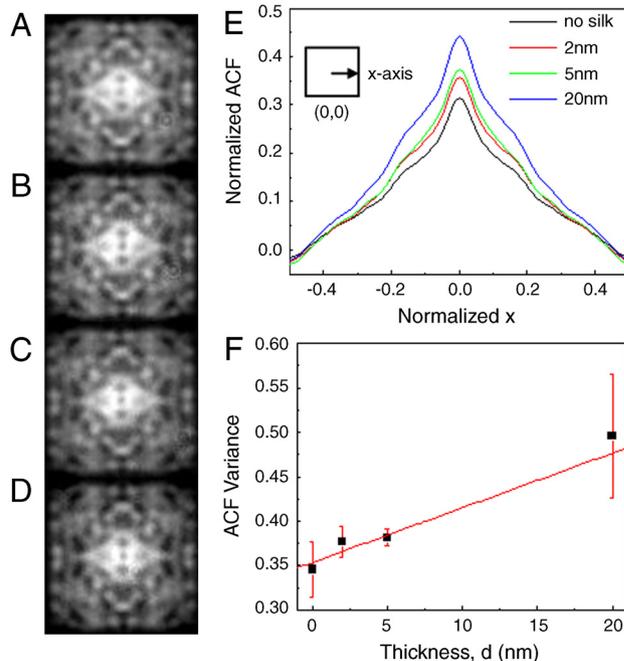
where  $s(x, y)$  is the fluctuating spatial signal and the angle brackets  $\langle \rangle$ , indicate averaging (integration) over the spatial domain. Once the normalized ACF of the structural color fingerprints of the aperiodic surfaces obtained from the bare and the silk coated surfaces has been calculated, the spatial modification of the fingerprints can be quantified by comparing their variances, which can be readily obtained by evaluating the normalization of the ACF in the limit of zero lateral displacements (29, 30):

$$\text{var } \delta s(x, y) = \lim_{\xi \rightarrow 0} \lim_{\eta \rightarrow 0} g(\xi, \eta) = g(0, 0). \quad [3]$$

It is important to note that this analysis, which is performed on the principal RGB spectral components of the scattered radiation, can unveil significant structural color modifications associated to the refractive index perturbation of aperiodic systems. In this work, we have demonstrated this effect by performing the autocorrelation analysis at the peak wavelength (622 nm) of the scattered spectrum of a Gaussian prime surface. The data shown in Fig. 5E, F summarize our findings. In Fig. 5E we plot the one-dimensional ACF profiles extracted from the two-dimensional intensity autocorrelation functions for different



**Fig. 4.** Colorimetric response of periodic gratings as a function of monolayer deposition. (A) shows the dark-field images of periodic gratings with no silk, 2 nm of silk, and 20 nm of silk (from top to bottom). (B, C) show the scattering spectral responses of the gratings, with lattice constant of (1) 600 nm, and (2) 700 nm correspondingly, coated with different thicknesses of silk protein monolayers. No protein detection can be observed in the 2–5 nm thickness range, while a small shift in the spectral peak is observed when 20 nm thick layers are deposited on the 700 nm grating (C).



**Fig. 5.** Autocorrelation analysis of structural pattern changes. Dark-field images corresponding to one spectral color (622 nm) of Gaussian prime lattice ( $\lambda = 300 \text{ nm}$ ) with different thicknesses of silk protein monolayers of (A) 0 nm (no silk), (B) 2 nm silk, (C) 5 nm silk, and (D) 20 nm silk under white light illumination were analyzed through (E) one-dimensional ACF profiles extracted from two-dimensional normalized autocorrelation function along the x-axis of the middle of the images. (F) The changes of patterns due to different thicknesses of silk protein monolayers are quantified by the normalized ACF variances.

thicknesses of the protein layer. The initial decay in the ACF reflects local short-range correlations in the aperiodic structure, while long-range correlations in the intensity pattern give rise to periodic oscillations in the ACF (30). The change in the structural color patterns (at any given wavelength of interest) induced by the presence of thin protein layers can be made quantitative by computing the variance of the scattered field intensity fluctuations. The experimental results in Fig. 5 indicate a substantial change in both the normalized ACF variance of perturbed colorimetric fingerprints and its complex spatial structure encoded in the ACF oscillatory behavior (which reflects the long-range oscillations). These data fully demonstrate the possibility of protein monolayer detection, in the visible spectral range, using conventional dark-field microscopy of aperiodic arrays of nanoparticles.

## Conclusions

In summary, by using frequency-resolved spatial analysis of colorimetric fingerprints in nanopatterned surfaces with deterministic aperiodic order we have demonstrated the ability to discriminate spectrally and spatially, in the visible spectral range, nanoscale surface variations down to the single (20 Angstrom) protein monolayer. The sensitivity levels are comparable to photonic crystals and surface plasmon biosensors making this a compelling alternative to these methods. Furthermore, the proposed detection scheme does not require dedicated setups and makes use of the conventional dark-field microscopy and standard image correlation analysis. The approach presented here introduces the general concept of multiplexed, label-free detection based on engineered colorimetric fingerprints in deterministic aperiodic structures.

These results, which can also be consistently obtained using other types of aperiodic surfaces, pave the way to the inexpensive, real-time sensing of protein monolayers in the visible spectral range using conventional microscopy techniques.

## Materials and Methods

**Gratings Fabrication.** Periodic and aperiodic nanoparticle arrays were fabricated using EBL on quartz substrates. The fabrication process flow is as follows: 180 nm of PMMA 950 (Poly Methyl Meth Acrylate) were spin-coated on top of quartz substrates, and the substrates were soft-baked

on a hot plate at 180 °C for 90 sec. 10 nm-thin continuous gold film was then sputtered on top of the resist to facilitate electron conduction for EBL writing. The nanopatterns were defined using a Zeiss SUPRA 40 VP SEM equipped with Raith beam blanker and Nanometer Pattern Generation System (NPGS) for nanopatterning. The resist was subsequently developed and a 40nm Cr thin film was deposited by e-beam evaporation. After lifting-off using acetone solution, the arrays with Cr nanoparticles were obtained. The resulting features of nanopatterned arrays are shown in Fig. S1 and are approximately 40 nm in height with radii of 100 nm, as measured by atomic force microscopy (AFM).

**Dark-Field Scattering Setup and Image Acquisition.** Fig. 1 B–D and Fig. 2F were collected in dark field under white light illumination using a backscattering microscope setup with a 50× objective (N.A. = 0.5) and a CCD digital camera (Media Cybernetics Evolution VF). The incident angle of the illumination was approximately 15° to the array plane, as shown in the Fig. 1E. Dark-field images and wavelength spectra were also measured in a transmission configuration using a dark-field condenser with N.A. 0.8–0.92. The transmitted light was collected with a 10× objective through a 1 mm iris (decreasing the N.A. ~0.1) and spectral images were obtained using a hyperspectral CCD (CRI Nuance FX) camera coupled to an Olympus IX71 microscope (Fig. 3, Fig. 4, Fig. 5).

**Silk Solution Preparation and Thin Film Deposition.** The silk solution was obtained by boiling *Bombyx mori* cocoons for 30 min in an aqueous solution of 0.02 M Na<sub>2</sub>CO<sub>3</sub>, then rinsed thoroughly with water to extract the glue-like sericin proteins, using methods we have previously reported. The solution was then dissolved in 9.3 M LiBr solution at room temperature, yielding a 20 wt % solution. This solution was dialyzed in water using a dialysis cassette with a molecular cutoff weight of 3,500 Da for 48 h. The silk fibroin solutions were poured onto nanopatterned quartz substrates and set to air dry in a laminar flow hood. The films were then left to dry for 24 or 48 h until all the solvent had evaporated to give solid fibroin protein silk films.

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# Supporting Information

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## SI Text

**Generalized Multiparticle Mie Theory.** We use the rigorous Generalized Mie Theory (GMT) approach (also called the rigorous theory of multipole expansions) (1–5) to provide a support and interpretation of our experimental data. Although the application domain of GMT is restricted to spherical scatterers, it yields the analytical solution of the scattering problem and results in highly efficient algorithms. In the frame of GMT approach, the electromagnetic field in a photonic structure of  $L$  nanoparticles can be constructed as a superposition of partial fields scattered from each particle. These partial scattered fields as well as the incident field and internal fields are expanded in the orthogonal basis of vector spherical harmonics represented in local coordinate systems associated with individual particles:

$$\mathbf{E}_{\text{sc}}^l = \sum_{n=1}^{\infty} \sum_{m=-n}^n (a_{mn}^l \mathbf{N}_{mn} + b_{mn}^l \mathbf{M}_{mn}), \quad l = 1, \dots, L. \quad [\text{S1}]$$

The use of the powerful addition (translation) theorem for vector spherical harmonics enables the transformation (translation) of the series expansion for the partial fields of the  $l$ -th particle into an expansion in the local coordinate system associated with any other particle of the array. A general matrix equation for the Lorenz-Mie multipole scattering coefficients ( $a_{mn}^l, b_{mn}^l$ ) can be obtained by imposing the electromagnetic boundary conditions for the tangential components of the electric and magnetic fields and by truncating the infinite series expansions to a maximum multipolar order  $N$ :

$$\begin{aligned} a_{mn}^l + \tilde{a}_n^l \sum_{j \neq l} \sum_{\nu=1}^N \sum_{\mu=-\nu}^{\nu} (A_{mn\mu\nu}^j a_{\mu\nu}^j + B_{mn\mu\nu}^j b_{\mu\nu}^j) &= \tilde{a}_n^l p_{mn}^l \\ b_{mn}^l + \tilde{b}_n^l \sum_{j \neq l} \sum_{\nu=1}^N \sum_{\mu=-\nu}^{\nu} (B_{mn\mu\nu}^j a_{\mu\nu}^j + A_{mn\mu\nu}^j b_{\mu\nu}^j) &= \tilde{b}_n^l q_{mn}^l \end{aligned} \quad [\text{S2}]$$

Here,  $A_{mn\mu\nu}^j, B_{mn\mu\nu}^j$  are the translation matrices, which only depend on the distance and direction of translation from origin  $l$  to origin  $j$  (1–4),  $\tilde{a}_n^l, \tilde{b}_n^l$  are the Mie scattering coefficients of  $l$ -th sphere in the free space (5); and  $p_{mn}^l, q_{mn}^l$  are the expansion coefficients of the incident field. Once truncated matrix Eq. S2 are solved for the scattering coefficients, the scattering, extinction, and absorption cross-sections as well as the scattered field distributions can be accurately calculated at any desired level of accuracy. The numerical solution of Eq. S2 can be obtained with a machine precision if the matrix equation is truncated at a high enough multipolar order.

**Image Correlation Analysis of Colorimetric Fingerprint.** We define the autocorrelation function (ACF)  $G(\xi)$  of a fluctuating spatial signal  $s(x)$  that describes the colorimetric fingerprint of nanoparticle arrays as:

$$G(\xi) = \langle s(x)s(x + \xi) \rangle, \quad [\text{S3}]$$

where the angle brackets  $\langle \rangle$ , indicate averaging (integration) over the spatial domain. In order to properly extract quantitative information, the spatial signal needs to be correctly normalized by defining the following quantity (6):

$$\delta s(x) = \frac{s(x) - \langle s(x) \rangle}{\langle s(x) \rangle} \quad [\text{S4}]$$

which enables proper definition of the normalized ACF:

$$g(\xi) = \langle \delta s(x)\delta s(x + \xi) \rangle = \frac{\langle s(x)s(x + \xi) \rangle - \langle s(x) \rangle^2}{\langle s(x) \rangle^2} = \frac{G(\xi)}{\langle s(x) \rangle^2} - 1. \quad [\text{S5}]$$

Analogously, for a colorimetric fingerprint in two spatial dimensions,  $s(x,y)$ , we can define the two-dimensional normalized ACF as:

$$g(\xi,\eta) = \langle \delta s(x,y)\delta s(x + \xi,y + \eta) \rangle = \frac{G(\xi,\eta)}{\langle s(x,y) \rangle^2} - 1. \quad [\text{S6}]$$

If the colorimetric fingerprint consists of an image with  $N \times M$  pixels, the discrete implementation of the spatially averaged ACF is readily obtained as:

$$g(\xi,\eta) = \frac{(1/NM) \sum_{k=1}^N \sum_{l=1}^M s(k,l)s(k + \xi,l + \eta)}{[(1/NM) \sum_{k=1}^N \sum_{l=1}^M s(k,l)]^2} - 1. \quad [\text{S7}]$$

We notice that this definition of normalized ACF enables us to obtain the variance of the spatial fluctuations of the colorimetric fingerprints by simple evaluation of the autocorrelation function in the limit when both  $\xi$  and  $\eta$  vanish (6):

$$\text{var } \delta s(x,y) = \lim_{\xi \rightarrow 0} \lim_{\eta \rightarrow 0} g(\xi,\eta) = g(0,0). \quad [\text{S8}]$$

In this paper, in order to perform the ACF calculations more efficiently, we have resorted to the well known Fourier transform relation (6, 7):

$$G(\xi,\eta) = F^{-1} \{ [F(s(x,y))] * [F^*(s(x,y))] \}. \quad [\text{S9}]$$

Once  $G(\xi,\eta)$  has been readily obtained from Eq. S9, we calculated the normalized ACF directly by using Eq. S6. The normalized ACF profiles in one spatial dimension (see Fig. 5 of the paper) are extracted from the two-dimensional normalized ACF along the center line ( $x$  axis) of the image and have been normalized with respect to the size of the array along the  $x$ -direction of the image.

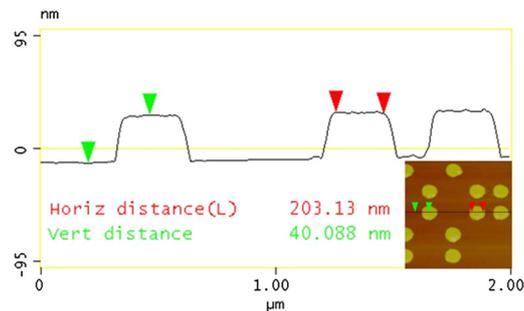
**General Principles of Linear Response Theory.** Our findings demonstrate that the colorimetric fingerprints of aperiodic structures with continuous spatial Fourier spectra are the most sensitive to small perturbation of the refractive index. This fact, which we have proved using full vector analytical Mie theory, can be more generally understood based on the general principles of linear response theory for stationary random signals. This theory provides the most general rationale to understand the scattering properties by rough surfaces in the linear optics regime. (We notice that the stationary hypothesis on the spatial signal (the scattering surface) is well satisfied in the limit of large samples). In fact, as long as the system's response is linear, we can express the mean square value of the system's output function  $E[y^2]$  (which in rough surface scattering corresponds to the scattered mean field fluctuations) as follows (8):

$$E[y^2] = \int_{-\infty}^{+\infty} |H(\omega)|^2 S_x(\omega) d\omega, \quad [\text{S10}]$$

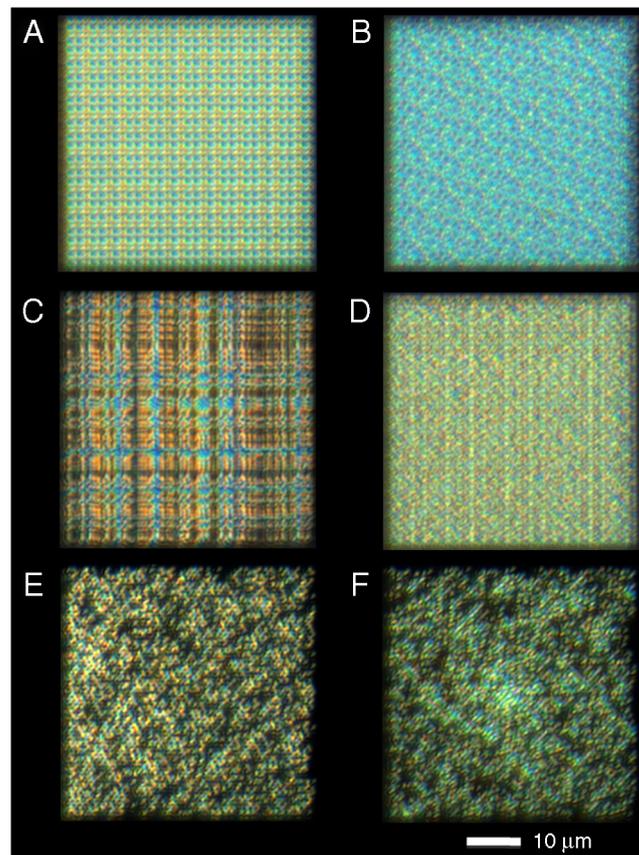
where  $H(\omega)$  is the linear optical transfer function of the system (frequency response),  $S_x(\omega)$  is the spectral density of the nanostructured surface (defined by the Fourier transform of its autocorrelation function), and  $\omega$  is a two-dimensional vector of spatial frequencies. It is evident from Eq. S10 that the spectral character, in particular the flatness of the spectral density, of aperiodic

arrays directly determines the intensity of the scattered field fluctuations. These fluctuations will be stronger for aperiodic arrays with “diffused” or flat Fourier spectra such as Rudin-Shapiro and Gaussian prime lattices. Therefore, Fourier space engineering of aperiodic arrays provides a simple tool for the optimization of the scattering response of deterministic aperiodic surfaces and allows to select the most appropriate surface structures to match specific application needs.

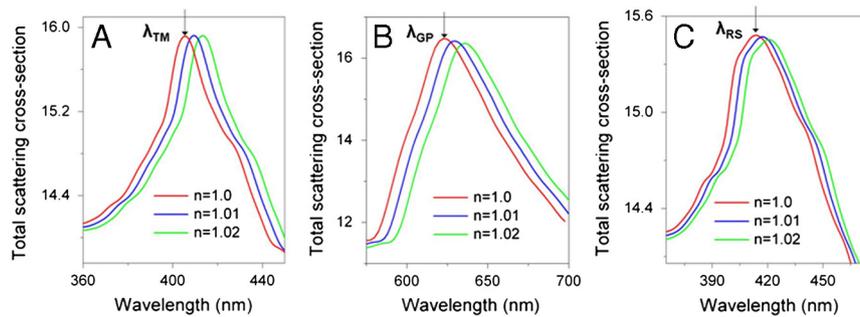
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**Fig. S1.** Atomic force microscopy (AFM) image of a Thue-Morse array with 40 nm-high, 100 nm-radius Cr nanoparticles, and minimum center-to-center interparticle separation of 400 nm.



**Fig. S2.** Dark-field scattering images of additional colorimetric fingerprints (A) Fibonacci, (B) Penrose, (C) Galois, (D) CoPrime, (E) Prime and (F) Ulam-Spiral aperiodic arrays of 100-radius and 40 nm-high cylindrical Cr nanoparticles on a quartz substrate.



**Fig. S3.** Calculated red-shifts of the peaks in the total scattering efficiency of (A) Thue-Morse, (B) Gaussian prime, and (C) Rudin-Shapiro aperiodic arrays of 200 nm-diameter Cr nanoparticles with the change of the ambient refractive index from  $n = 1.0$  (red) to  $n = 1.01$  (blue) to  $n = 1.02$  (green). The nearest center-to-center interparticle separation is 300 nm for the Gaussian prime array and 400 nm for Thue-Morse and Rudin-Shapiro arrays. The arrows indicate the wavelengths of the resonant peaks in the scattering spectra of the arrays in air.

